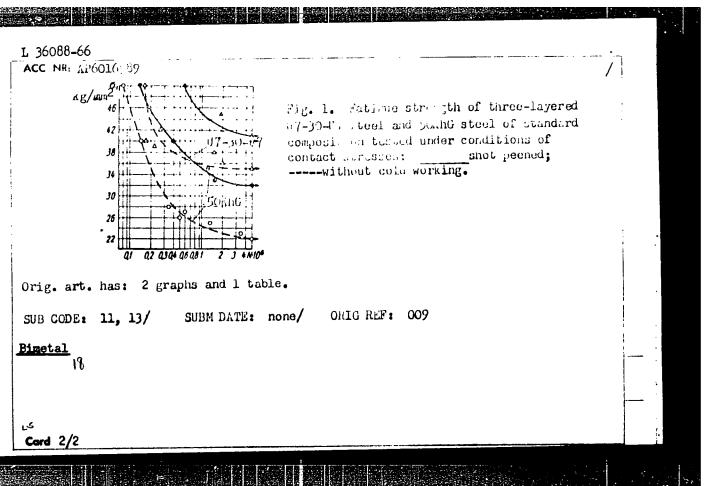
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L 36088-66 EWT (m) / EWP (w) / T/EW ACC NR: A. 6016589 (A, N))00/005/002 7 /0029
A.M.SRS: Lozinskiy, M. G.; Tem	iyanko, V. G.; Hatanzon, Ye. I.	41
ONG: Institute of Mechanical En	ngineering (Institut mashinostroyeniy	
TITLE: The use of three-layered	U7-30-U7 steel for automobile spring	
SOUIGE: Mctallovedeniye i termi	cheskaya obrabotka metallov, no. 5, 1	1966, 27-29
TOPIC TAGS: contact stress, sho steel, 30 steel, 50KinG spring st	ot peening, fatigue strength, spring seel	steel, steel/U7
string are given. This material strikes with le properting a duct to in outer layers of U70steel and spring band with a thickness of the ers of 17 steel is 1.2 mm, the recording when the total the subscript are found to have a himmesses than 10khG steel. The	ly of three-layered U7-30-U7 steel for all was used to obtain high hardness of the center. The material consists of a center layer of 30 steel. In ord 7 mm after rolling when the thickness of the blank of U7 steel ickness of the packet is 180 mm. Springler fatigue limit under the influence fatigue strength of the three-layered 3-30% greater than that of 50KhG steel.	t the spring-leaf from mratively der to obtain a softhe outer must be 30 mm rings of this e of contact from the steel that
Card 1/2	UDC: 621.1	135.3:621.771.8



ACC NR: AP7003847

(A,N)

SOURCE CODE: UR/0122/67/000/001/0057/0059

AUTHORS: Sveshnikov, D. A. (Engineer); Natanzon, Ye. I. (Candidate of technical

ORG: none

TITLE: The possibility of replacing alloy steels with carbon steels case-hardened by heating with high-frequency current

SOURCE: Vestnik mashinostroyeniya, no. 1, 1967, 57-59

TOPIC TAGS: alloy steel, carbon steel, case hardening, electric current, metal stress, hardness, fatigue strength, durability/ 45 carbon steel, 35KhGSA alloy steel, 40KhGTR alloy steel

ABSTRACT: The results of a study involving the general laws governing the distribution of residual stresses induced during case hardening by heating with high-frequency current are presented as functions of the depth of the hardened layer and the temperature of subsequent tempering. Specimens of 45 steel with an outside diameter of 80 mm and a wall thickness of 7.0 mm (tubes) were used. The obtained results show that case hardening causes residual compressive stresses averaging 75 kg/mm² on the surface (see Fig. 1). It was determined that subsequent tempering is not advisable in a number of cases. Case hardening was found to be an effective method for increasing residual

Card 1/2

VDC: 621.785.616:621.3.023:669.14

ACC NR. AP7003847

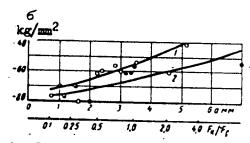


Fig. 1. Residual stresses in the surface layer of rings (at depth of 0.1 mm) versus depth of hardened layer and ratio F_k/F_c : 1 - HRC 55; 2 - HRC 45

compressive stresses which increase the fatigue strength and durability of casehardened and tempered parts. Engineers L. D. Golubovskaya and A. I. Bad'in took part in the work. Orig. art. has: 5 graphs.

SUB CODE: 11, 13/ SUBM DATE: none/ ORIG REF: 002/ OTH REF: 002

Card 2/2

MATAPOV, B. 5.

Mor. Central Laboratory of the Plant imeni Baranov (-1945-)

Candidate of Technical Sciences

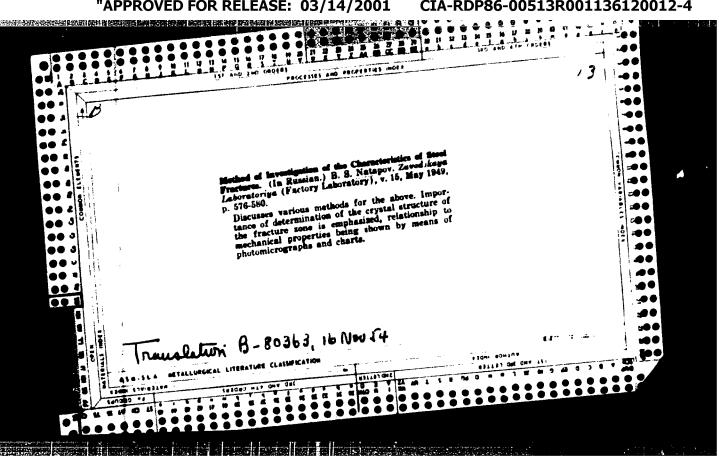
"Increasing the Morkability of Allcy Case-hardened Steel, Stanki T Instrument, 16, No. 9, 1945

BR-52059019

WATAPOV, B.S., kandidat tekhnicheskih nauk.

Crysteline frecture in E172 steel; its nature and prevention.
Stal' 7 no.3:245-248 '47. (MEMA 9:1)

1.Exporoshskiy svtomekhanicheskiy institut.
(Steel--Defects)



NATAPOV, BORIS SOLOKONOVICH

N/5 615.918 .N2

Termicheskaya Obrabotka Metallov (Heat Treatment of Metals, by) B. S. Solomonovich i <u>Mikolay Arkad'yevich Blagoveshchenskiy</u>. Moskva, Metallurgizdat, 1955.

392 p. Illus., Diagrs., Tables. Literatura: p. 391-392.

WATAPOV, Boris Solomonovich; LAKHTIN, Yu.M., redaktor; GOLYATKINA, A.G., redaktor izdatel'stva; EVENSON, I.M., tekhnicheskiy redaktor

[Metals] Metallovedenie. Moskva, Gos. nauchno-tekhn. izd-vo lit-ry po chernoi i tevetnoi metallurgii, 1956. 343 p. (MIRA 9:12)

(Metals)

60V/133/58-9-20/29

AUTHORS: Natapov, B. S. and Tsivirko, D. Ye.

The Influence of Structure and Mechanical Properties of TITLE: Steel O8kp on Its Stamping Ability (Vliyaniye struktury i mekhanicheskikh svoystv stali O8kp na yeye shtampuyemost')

PERIODICAL: Stal', 1958, Nr 9, pp 828-834 (USSR)

ABSTRACT: The investigation of the above problem was carried out in order to establish the relationship between the results of testing under laboratory conditions of physico-mechanical properties and micro structure of sheet steel O8kp and its actual behaviour during the stamping process. A statistical analysis of defective stampings of parts of motorcar bodies (at the Gor'kiy Automobile Plant) due to tearing was made and correlated with properties of metal. Correlation coefficients of a number of defects during stamping with physico-mechanical properties of steel -Table 1 and Fig.1; the influence of the initial mechanical properties of steel on the number of defective stampings - Table 2, and the chemical composition of respective metal deliveries in the ladle and in sheets - Table 3; frequency of distribution of mechanical properties of the individual deliveries of sheets - Fig. 2; the comparison of the proportion of defects during stamping Card 1/4 with mechanical properties of metal - Table 4; frequency

Sov/133/58-3-20/29

The Influence of Structure and Mechanical Properties of Steel O8kp on Its Stamping Ability

distribution of the degree of deformation for sheets of the same thickness - Fig.3. It is concluded that: 1) Under correct technological conditions of stamping parts of motor car bodies the proportion of rejects caused by faults in the metal amounted to 70-80% of the total and due to deformation 20-30% (mainly due to incorrect pressing technology). 2) One of the main causes of tearing due to faults in the metal are laminations as well as the following deviations of methanical properties from optimum values: a) increased thoughness of metal ($\sigma_s > 22 \text{ kg/mm}^2$, the ratio of $\sigma_s/\sigma_B > 0.75$, $\sigma_s > 0.75$,

ions in zones of maximum deformations (at the size grain 6.6-7.7 and 8 and any combinations of the above sizes; round Card 2/4

Sov/133/58-9-20/29

The Influence of Structure and Mechanical Properties of Steel 08 kp on Its Stamping Ability

cementite inclusions of the size 1-5 are permissible).

4) Grading of rejects on works according to tears during stamping should be carried out after an investigation of fractures in the zone of dangerous deformation and according to additional mechanical tests of sheets before stamping.

5) Limiting permissible indices of mechanical properties for each part, difficult to stamp, should be established by a cooperative investigation of metallurgical and motor car works. 6) For the control of stability of the technological process of production of sheet steel tensile tests or according to Eriksen and hardness tests. 7) In order to improve stamping ability of sheet steel it will be advantageous to decrease the yield point (ratio of oS/oB) to increase the relative elongation, to secure freedom from laminations

Card 3/4

Sov/133/58-9-20/29

The Influence of Structure and Mechanical Properties of Steel Ockpon Its Stamping Ability

and to increase the uniformity of sheets in a delivery. There are 4 tables, 3 figures and 12 references, of which 9 are Soviet and 3 English.

ASSOCIATION: Zaporozhskiy mashinostroitel'nyy institut (Zaporozh'ye Machine Building Institute)

Card 4/4

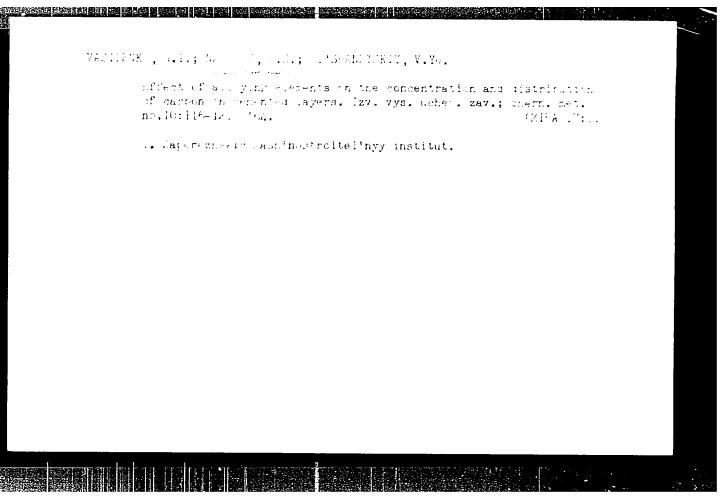
HATAPOV, B.S., dots., kand.tekhn.nauk; FAL'KEVICH, E.S., inzh.

Effect of gases on the bardness of rimmed and killed low-cerbon steels. Izv.vye.ucheb.zav.; chern.met. no.11:95-98 N '58.

(MIRA 12:1)

1. Zaporozhskiy mashinostroitel'nyy institut. Rekomendovano kafedroy tekhnologii metallov i metallovedeniya.

(Steel--Metallurgy) (Gases in metals)



SOV/32-24-6-32,43 Natapov, B. S., Fal'kevich, E. S. AUTHORS:

The Determination of Mechanical Properties of Steel by TITLE:

Testing Its Ability to Be Cupped as Indicated by Coercive Force Values (Ob opredelemii mekhanicheskikh svoystv stali dlya

glubokoy vytyazhki po znacheniyam koertsitivnoy sily)

Zavodskaya Laboratoriya, 1958, Vol. 24, Nr 8, PERIODICAL:

pp. 1013 - 1014 (USSR)

The basic properties of the quality of cold-rolled, cupped ABSTRACT:

steel plates were found to be: the flow limit, the relation-

ship

σs the hardness, and the results of the Eriksen test (Ref 1)

The existing testing methods to determine flow limit and hardness are inadequate when applied to steel plates, so investigations were undertaken to determine whether these

values could adequately be determined by the method of

coercive force measurement. The method worked out by S.A. Saltykov

(Ref 4) was among those used in these determinations. The

experimental results obtained show that with a change of Card 1/2

SOV/32-24-8-32/43 The Determination of Mechanical Properties of Steel by Testing Its Ability to Be Cupped as Indicated by Coercive Force Values

the coercive force the changes in the flow limit of and the hardness $R_{\underline{B}}$ depend on the particle size and on the grade of trimming. These relationships are expressed in the following equations:

 $\sigma_{S} = 14 H_{C} + 1.5$; $R_{B} = 20 H_{C} + 10$. Students Ye.P.Ponomarenko. V.G.Steshenko, and K.K.Milyayev participated in the work. which are Soviet There are 3 figures and 4 references

ASSOCIATION: Zaporozh skiy mashinostroitel'nyy institut (Zaporozh'ye Institute for Machine Construction \

Card 2/2

18(3), 18(7)

AUTHORS: Natapov, B. S., Vasilenko, G. I., Ryabtsev, S. I., Panasenko, Ye. I.

807/163-59-1-43/50

TITLE:

Influence of Hot Rolling and of Recrystallization Annealing Upon the Structure and the Properties of Steel O8kp (Vliyaniye goryachey prokatki i rekristallizatsionnogo otzhiga na

strukturu i svoystva stali 38kp)

PERIODICAL:

Nauchnyye doklady vysshey shkoly. Metallurgiya, 1959, Nr 1,

ABSTRACT:

This is an investigation of some regularities encountered in the formation of the structure of steel O8kp during hot rolling and annealing. The samples were rolled in laboratory rolling mills and annealed in muffle furnaces. The structure was investigated by means of micrograph and X-ray analysis methods.

The grain sizes were qualitatively determined on the

texturometer by Akulov. In the plant "Zaporozhstal' " steel sheet has been hot rolled in order to obtain a high degree of stretching on a continuous rolling mill with 4 stands of

roughing rolls and 6 stands of dressing rolls. The investigation

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showed that if the temperature after rolling still exceeds 880° recrystallization in the billet takes place within a very

Influence of Hot Rolling and of Recrystallization Annealing Upon the Structure and the Properties of Steel O8kp

SOV/163-59-1-43/50

short time (below 1 second). The crystal ization at 1 5500 starts not before 16 seconds. Immediately after leaving the last stand of rolls the metal exhibited a fine grain (index 7-3) at all temperatures. Grain growth starts only after an appropriate halting time, for example at 800° growing begins after 1 minute (Ref 1). If rolling is terminated at a temperature of 860-840° and if the billet is kept for 30 minutes in a furnace at the same temperature no grain growth is found. A pronounced growth of the grains at the surface of the semifinished steel products occurs at a temperature after rolling which is below the critical point Ar_{χ} , at a reduction of less than 15% and a subsequent halting time of 15-30 minutes at 800-750°. When rolling is carried out in the plant, a difference is observed in the growth of the grains at the metal surface. This is considered due to the different depth of deformation and a slow cooling in the range of 800-700°. In order to prevent the formation of eutectic grains the temperature at the end of rolling must exceed 860°, the reduction must keep

Card 2/4

Influence of Hot Rolling and of Recrystallization Annealing Upon the Structure and the Properties of Steel 08kp

SCV/163-59-1-43/50

within the limits of 12-20% and the cooling process should be accelerated in the range of 800-680°. It was further found that under otherwise equal conditions the grain size after rolling depends upon the chemical composition and the metallurgical history of the melt. The investigation showed that the duration of annealing of the steel O8kp can be reduced and the rate of cooling can be increased. One of the means of a considerable increase in output and of an improvement in steel quality may be a continuous electric annealing of the steel sheet. Laboratory investigations of resistance heating conducted in this direction yielded a satisfactory structure and quality of steel O8kp. Two temperature ranges were found to be most suitable: 730-780° and 1,000-1,050°. An annealing at 730-780° is most advantageous with respect to economy and production considerations. There are 5 figures, 2 tables, and 5 references, 4 of which are Soviet.

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ASSOCIATION: Zaporozhskiy mashinostroitel'nyy institut (Zaporczh'ye Institute of Machine Building)

30V/163-59-1-49/50 18(3)

Natapov, B. S., Tsivirko, D. fe., Fal'kevich, E S. AUTHORS:

Influence of Several Factors Upon the Quality of Automobile Plate .TITLE: (Vliyaniye razlichnykh faktorov na kachestvo avtomobil'nogo lista)

Nauchnyye doklady vysshey shkoly. Metallurgiya, 1959, Nr 1, PERIODICAL:

pp 254-258 (USSR)

In order to determine the deep drawing quality of O8kp VGV steel ABSTRACT:

sheet the influence of the physical and mechanical properties upon the quality of the products was investigated by making a direct study of the deep drawing process for which methods of statistical analysis were applied (Refs 1,2). This statistical analysis rested on the tests of the physical and mechanical properties of 630 lots of the steel sheet in question. The experiments were carried out in the laboratory of the Gor'kovskiy avtozavod (Gor'kiy Automobile Works). This article presents the results of the statistical analysis in a table exposing the dependence of the percentage of substandard products upon the characteristics of the metal. A linear relationship was found to exist between the amount of substandard products caused by fissures in the sheet, and the physical and mechanical properties. The information collected in-

dicates that none of these properties exercises a dominating Card 1/3

30V/163-59-1-49/50

Influence of Several Factors Upon the Quality of Automobile Plate

influence. A relatively insignificant interdependence between the amount of substandard products in deep drawing and the grain size of the ferrite and the inclusions containing structureless zementite was established. The low correlation coefficient is explained. The results of these investigations permit to make the statement that metallurgists should strive to produce a metal with uniform properties through one sheet and through one series and with a minimum number of cracks. As hitherto only very insufficient methods have been known of determining the yield point and the hardness the authors tried to investigate these characteristics by measuring the coercive force and thus found out that these two properties take a largely parallel course. Hence the usual measuring methods can be supplanted by a measurement of the coercive force. The mechanical properties of steel sheet can be improved by a larger grain size, by removing non-metallic inclusions, by a prevention of zementite formation and by a reduction of the sulphur content. The annealed sheets are dressed in order to improve the surface quality of automobile parts. By the specifications of the "Zaporozhstal' Works the reduction in height by dressing is set to 0,8 - 1.5 %. The investigation of the rules governing the aging of O8kp steel after deformation showed that the hardness during aging increases the

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SOV/163-59-1-49/50

Influence of Several Factors Upon the Quality of Automobile Plate

more, the more nitrogen is contained in the steel An increase of the aluminum content in the steel leads to a reduction of hardness during aging. A steel which has been deoxidized by aluminum shows during aging towards aging if it contains ample amounts of oxygen a tendency towards aging if it contains ample amounts of oxygen. The microscopical investigation of samples destroyed at the temperature of liquid nitrogen showed that by aging the bonds at the grain ture of liquid nitrogen showed that by aging the bonds at the grain boundaries are weakened. There are 1 table and 5 Soviet references.

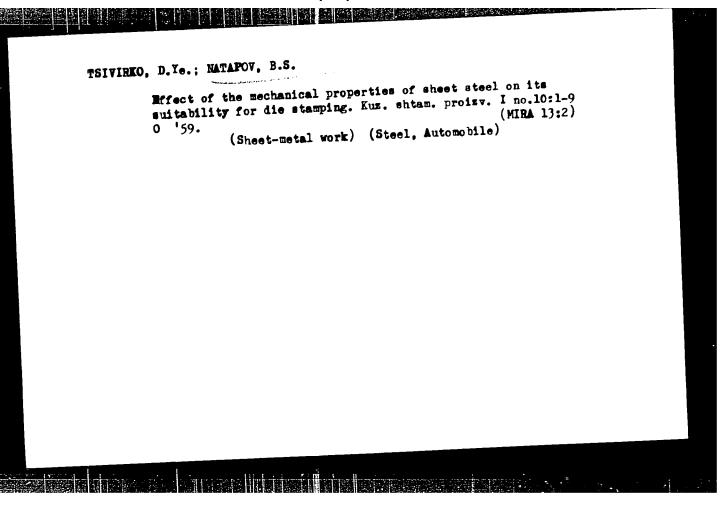
ASSOCIATION:

Zaporozhskiy mashinostroitel'nyy institut (Zaporozh'ye Institute of Machine Building)

SUBMITTED:

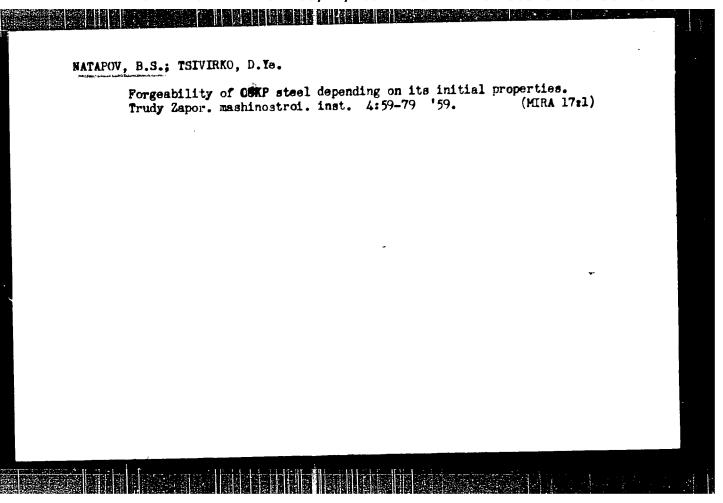
May 4, 1958

card 3/3



NATAPOV, B.S.; VOLOSHCHUK, M.D.; LEVCHENKO, T.V.; TSIVIRKO, D.Ye.

Dependence between the mechanical properties and the microstructure of OSIP steel. Trudy Zapor. mashinostrol. lnst. 4:45-58 '59. (MIRA 17:1)



S/148/60/000/006/014/016/XX A161/A030

AUTHORS: Natapov, B.S.; Fal'kevich, E.S.

TITLE: Mechanical Aging of 08km (08kp) Steel

PERIODICAL: Izvestiya vysshikh uchebnykh zavedeniy. Chernaya metallurgiya, 1964

No. 6, pp. 114 - 118

TEXT: The Zaporozh'ye Institute of Machine Construction has studied the mechanism of mechanical aging of O8kp automobile sheet steel, i.e., the change of mechanical properties with the time after dressing. The chemical composition of the samples from 14 steel consignments conformed with the standard (GOST) requirements. Dressing was produced on a laboratory mill with 1 to 25% reduction; tempering in 250°C for one hour was used after dressing. Besides this, the effect of dressing and rolling was studied in a rolling shop on two heats of rimming O8kp and two heats of semi-killed O8mc (O8ps) steel of the "Zaporozhstal'" Plant. Dressing with 0.1 - 1.5% reduction produced the best effect. The intensity of mechanical aging was lower in semi-killed steel than in rimming. The difference of mechanical properties before and after rolling was not high. Slip lines appeared in the stamping of sheets reduced 0.5 - 1.0% in dressing; no slip

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Mechanical Aging of 08κπ (08kp) Steel

S/148/60/000/006/014/016/XX A161/A030

lines formed in sheets dressed with a reduction over 1%. As had been stated previously (Ref. 4), the increase of hardness in the mechanical aging of O8kp steel is connected with the nitrogen content. Oxygen and hydrogen have no such effect. It was not possible to eliminate the effect of mechanical aging fully by additions of aluminum (Ref. 5) and it is apparent that carbon caused the aging of samples deoxidized with aluminum. The mechanism of mechanical aging was studied with an 3M-3 (EM-3) electronic microscope; x-ray structure analysis by the Darvin (Ref. 6) and L.I. Lysak (Ref. 7) methods; the 2nd kind distortions were determined by Lysak's method (Ref. 7), and the 3rd type of distortions by a modified method of A. Kokhanovskaya (Ref. 8). An YPC-50 (URS-501) apparatus was used for the x-ray analysis. Natural mechanical aging during 8 days did not change the blocks dimensions and the 2nd kind of stresses (distortions), but the 3rd kind of stresses diminished. Aging at a raised temperature increased stresses and caused the blocks to split. The electronic microscope revealed no phase changes at natural aging, but artificial mechanical aging was accompanied by a segregation of the particle on the grain boundaries and the appearance of small boils on the slip lines (Fig. 3). The boils appeared to be nitrides and carbides (their nature could not be determined). The experiment data meet the dislocation theory of the mechanical aging process. According to this theory, the first stage of the proc-

Card 2/4

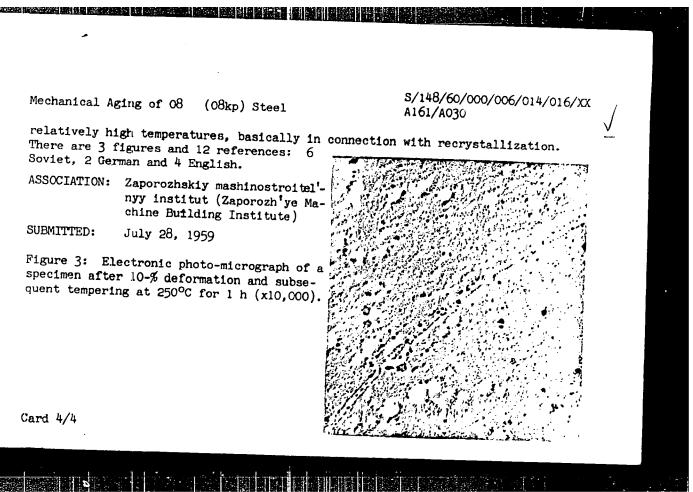
Gechanical Aging of 08 (08kp) Steel

S/148/60/000/006/014/016/73. A161/A030

ess consists in the migration of nitrogen and carbon atoms to dislocations, and the formation of "clouds" around the dislocations. Accumulating in spots of maximum tension stresses (below the dislocation center), they must partly release the stresses around the dislocations, and this explains the changes of 3rd kind of stresses in natural aging. The first aging stage is not connected with the separation of a second phase and with plastic deformation, and this accounts for the lacking blocks splitting and growth of the 2nd kind of stresses. Judging by the data of the papers (Refs. 10 - 12), mechanical aging at raised temperatures must end with the formation of subdispersed particles, and the observations in the present investigation confirmed this assumption. It follows that the nature of the mechanical aging may be explained with migration of nitrogen and carbon atoms to dislocations and the subsequent formation of "clouds". The "clouds" block the dislocations, and this results in an increased yield limit, tensile strength, hardness and other properties. All these properties are connected with the displacement of dislocations. The process speeds up with an increase in temperature, and subdispersed segregations appearing later in dislocations are firmly bound with the dislocations themselves. The firm bond of the nitrogen and carbon atoms with the dislocations obstructs the coagulation processes. This explains the peculiar phenomenon of restoring physical and mechanical properties at

V

Card 3/4



HIPHP.

\$/129/62/000/005/009/009 E193/E385

1,1710

AUTHOR: Natanov B.S., Candidate of Technical Sciences

TITLE Heat-treatment of case-hardening steels IEXHBA(lokh)VA)

and 12×244A (12Kh2N4A)

PERIODICAL. Metallovedeniye i termicheskaya obrabotka metallov no. 3, 1962 50 - 59

TEXT: The theoretical basis of the present paper is provided by data reproduced in a figure which shows the remperature-time-transformation (TTT) curves for case-hardening steels lokn. A. 12Kh2N4A and 12XHBA (12KhNVA) (diagrams a 5 and 3. respectively), the continuous and broken lines relate respectively to the case and core carburized test pieces. After discussing the significance of these diagrams, the author proceeds to give recommendations regarding the optimum heat treatment procedures. It is pointed out that whereas steels used in responsible carburized components should have FTS of

95 - 115 kg/mm 2 , the hardened steels 12Kh2N+A, 18KhXVA and 12 \times 45A (12KhN5A) with a carbon content near the upper limit of

Card 1/5

\$/129/62/000/005/005/005/009 E193/E383

Heat-treatment of . .

Card 2/b

of the specified composition have U(S of $120 + 150 \text{ kg/mm}^2$ which, apart from reducing their impact strength makes them difficult to machine The machineability of carburized components can be improved by application of martempering or combined quenching as the final heat-treatment. Since the stability of austenite of the carburized layer at 500 - 250 °C is higher than that of the austemite of the core it is possible to cool carbunized components an the isothermal medium in such a way that the holding time is sufficiently long to convert the austenite of the core into martensite and vet short enough to prevent transformation in the carburized layer. Martempering of steel lokhNVA at 550 - 450 C is not recommended because treatment at these temperatures does not reduce hardness of the core owing to the high stability of austenite. The same treatment applied to steel 12KhN3A brings about a decrease in its UTS but its impact strength is also reduced (the latter effect is attributed to the formation of a heterogeneous structure containing isolated zones of territe and troostite or even martensite! However, martempering of steel 18KhNVA at lower temperatures

S/129/62/000/003/009/009 E193/E383

Heat-treatment of

brings about both a decrease in hardness and an increase in impact strength. The recommended treatment consists of heating the component in a salt bath or an electric furnace to 800 - 830 °C; cooling it in a nitrate bath maintained at 250 holding at that temperature for 10 - 15 min and quenching in oil or water. In this way, satisfactory values of hardness of both the core and the case are attained, the degree of distortion being less than that caused by the conventional treatment. When the carbon content of steel 18KhNVA is below 0.18%, satisfactory hardness (HRC 39-34) of the core of carburized components can be attained by cooling in air from 820 - 840 °C, however, it is impossible to attain hardness RC 60 in the carburized layer with this treatment. The carburized layer will have hardness RC 60 after quenching in oil but the hardness of the core (RC 41) will be too high. The core hardness in carburized components of such steels cannot be lowered by austempering and in this case martempering is recommended (fast cooling in oil or a nitrate bath in the upper range of sub-critical temperatures (650 - 500°C) and slow cooling in the lower range (450 - 150 °C)). The

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Heat-treatment of

S/129/62/000/003/009/009 E193/E383

remaining unaffected;

3) cooling to convert the austenite of the carburized layer to martensite. The following schedules can be recommended. For steel $18 \mathrm{KhNVA}$ - heating to 820 - 840 °C, cooling to 260 - 330 °C in air, oil or a nitrate bath, heating to 560 - 580 °C, holding at that temperature for 1 - 3 hours and quenching in oil. For steel $12 \mathrm{Kh2N4A}$ - heating to 840 - 850 °C, cooling to 260 - 230 °C in oil or a nitrate bath, heating to 500 °C, holding at the temperature for no longer than 1 hour and quenching in oil. The effectiveness of this treatment which can be used for steels, whose austenite is sufficiently stable at 450 - 600 °C, has been confirmed experimentally.

Card 5/6

NATAPOV, R.S., kand.tekhn.nauk

Anomaly of the structure of a cemented layer. Metalloved. i term. obr. met. no.6:23-27 Je *62. (MIRA 15:7)

1. Zaporozhakiy mashinostroitel nyy institut. (Steel-Metallography) (Case hardening)

S/126/62/013/006/016/018 E071/E192

AUTHORS: Natapov, B.S., and Ol'shanetskiy, V.Ye.

TITLE: On the coalescence of carbide phase in normal and

abnormal carbon steels

PERIODICAL: Fizika metallov i metallovedeniye, v.13, no.6, 1962,

A CONTROL OF THE RESEARCH OF THE PROPERTY OF T

934-936

TEXT: The velocity of coagulation of cementite grains is usually related to the velocity of diffusion of carbon. However, K.P. Bunin considered that the limiting factor in the kinetics of the coagulation process is the diffusion migration of vacancies and not the velocity of diffusion of carbon in α- and γ-phases. In order to elucidate the influence of the velocity of diffusion of carbon on the process of coalescence of cementite, specimens of normal (08 kΠ) (08kp) and abnormal (non-ageing steel 08 H) (08Yu) steel possessing different coefficients of diffusion of carbon in ferrite were taken. The specimens were submitted to cementite treatment at 950 °C for 10 hours, then hardened in water from 970 °C in order to obtain martensite structure. Hardened specimens were isothermally treated at 550 °C and 700 °C for Card 1/2

HATAPOV, B.S.; BARZIY, V.K.; OL'SHAMETSKIY, V.Ye.; Prinimali uchastiye: FILOMOV, V.A., insh.; TUDIM, M.I., insh.; IOFFE, M.M., insh.; FOPOV, S.M., insh.; RYBALKO, G.I., insh.; ODIMETS, L.I., insh.; SIGALKO, F.V., insh.; TSIVIRKO, D.Ye.; VOLOSHCHUK, M.D., insh.

Heat treatment of cold-rolled sheet metal. Stal' 22 no.2:163-165 F '62. (MIRA 15:2)

1. Zaporozhskiy mashinostroitel'nyy institut i zavod
"Zaporoshstal". 2. Zavod "Zaporoshstal" (for Filonov,
Tudin, Ioffe, Popov, Eybalko, Odinets). 3. Zaporoshskiy
mashinostroitel'nyy institut (for Sigalko, TSivirko, Voloshchuk).

(Sheet steel—Heat treatment)

S/133/63/000/001/010/011 A054/A126

AUTHORS:

TITLE:

Natapov, B. S., Soroko, L. N., Barziy, V. K., Filonov, V. A. (Deceased), Gurskiy, G. L., Ioffe, M. M., Letchford, N. I., Yudovich, S. Z.

٠.

Improving the stamping properties of O8 N (O8Yu) grade sheet steel

PERIODICAL: Stal', no. 1, 1963, 84 - 86

TEXT:

A new technology has been developed to produce low-carbon (0.04 - 0.08%) steel suitable for cold rolling of automobile sheets having good stamping properties and which do not tend to age. From the tests (carried out in co-operation with I. A. Goncharov, G. Mikhaylov, F. A. Ksenzuk, V. G. Antipenko, M. Ye. Kugayenko, L. Dobrovol'skiy, L. I. Odinets, N. P. Cherkashina, A. K. Yaitskiy, I. N. Avramenko, M. I. Lyakhova, R. I. Razumovskaya, S. M. Popev, A. L. Khudas ("Zaporozhstal'"), N. P. Semperovich, V. Ye. Ol'shanetskiy, M. D. Voloshchk, F. V. Sigalko (ZMI), K. M. Romanycheva, V. G. Kochevatov (GAZ)) it was concluded that the manganese content of the test grade should be lowered to 0.24 - 0.35%, while the quantity of other elements that increase the hardness

Card 1/2

s/133/63/000/001/010/011 A054/A126

Improving the stamping properties of...

of the steel (C, N, Si, Cu, etc.) should also be kept as small as possible. content of residual aluminum, which has a stabilizing effect, should be inwreased to 0.04 - 0.09% (1. e. 900 - 1,100 g/ton in the mold), the temperature at the end of rolling should be 850 - 920°C, the winding temperature after rolling 540 - 610°C, which promotes the formation of oblong ferrite grains and improves the cementite distribution. The finishing stand should be adjusted to reductions of 0.6 - 1.8%. The new steel is suitable for very deep drawing (according to COCT 9045-59 (GOST 9045-59). In the tests aluminum of a purity of 99.9% and another kind having 13% admixtures were used. However, the favourable results obtained with the 99.9% aluminum could only be approximated, but not achieved with the second grade aluminum, even when in the latter case the annealing time was extended from 8 to 12 hours. There are 1 figure and 2 tables.

ASSOCIATION: Zaporozhskiy mashinostroitel'nyy institut (Zaporozh'ye Engineering Institute), Zavod "Zaporozhstal" (Zaporozhstal" Plant), and Gor'kovskiy avtomobil'nyy zavod (Gorkiy Automobile Plant)

Card 2/2

L 10600-63 MAP(q)/BMT(m)/BDS AFFEC/ASD JD

ACCRESION NEL AP3001053

8/0148/63/000/004/0115/0123

AUTHER: Matapov, B. S.; Ol'shanetskiy, V. B.; Vasilenko, G. I.; Voleshchuk, M. D.

TIME; The mechanism of normal and abnormal steel structure formation

Source: 1707. Chorneys metallurglys, no. 4, 1963, 115-123

TOPIC TAGS; absenual steal structures, structural transfermation, hyperentectoid

ABSTRACT: The study was made in order to explain the formation of an abnormal structure in steels, and to what extent the surface energy influences the rate of independent or occuperative growth of different structural components. Samples of normal and abnormal steel of type OSkp with chemical composition C Si Mm P S Al. and cast at the Zaporoshstal Works, were carbonised for 10 hours in bondirizing carboniser at a temperature of 930C. In order to observe the structural transformation in steel, the samples were heated to 900C, then submerged at certain temperatures in a salt solution (50% ECI plus 50% ECI) and then quenched in water. After the hyperentectoid steel to a point above A sub cm and at subsequent isolating the hyperentectoid steel to a point above A sub cm and at subsequent isolatics began to take place. With an extended duration, this lattice of cemencial computation in the nermal steel. In the abnormal steel an intensive

L 10600-63

ACCESSION IN: AP3001053

dealescence of community takes place. In order to explain the differences between normal and absermal steels, the isothermic transformation of austenita at different temperatures was studied by annealing and subsequent study of the microstructure of the samples. The abnormal structure in the steel is formed as a result of the precediment separate growth of phases, assuming, that in this process the decisive factor is the ferrite crystallization rate. The formation of an abnormal structure is observed in both the normal and the abnormal steel when the austenite is superscelled to a point just below A sub 1. The tendency to form an abnormal structure in steel is greater, when the surface tension at the boundaries of ferrite-austenite and community-austenite is lower. Orig. art. has: 5 figures and 1 table.

ASSOCIATION: Zaporezhakiy Mashinostroitelny*y institut (Zaporezh Machine-Building

30 MITTED: 254pr62

DATE ACQD: 11Jun63

ENCL: 00

SEE CODE: 00

NO REF SOV: 014

OTHER: 005

Care 2/2

NATAPOV, B.S.; OL'SHANETSKIY, V.Ye.; VASILENKO, G.I.; VOLOSHCHUF, M.L.

Effect of various factors on the tendency of steel towards anomalies. Izv. vys. ucheb. zav.; chern. met. 6 no.8:141-150 '63. (MIEA lo:11)

1. Zaporozhskiy mashinostroitel'nyy institut.

AIRKSEYENKO, M.F.; BANAS, P.S.; BOBKOV, T.M.; NATAPOV, B.S.; RYABTSEV, S.I.;
SKLYAROV, P.I.; FRANTSOV, V.P.; YUDOVICH, S.Z.; PRONIN, V.Ie.

DI-1 stainless steel. Stal' 23 no.2:159-162 F '63. (MIRA 16:2)
(Steel, Stainless)

L-36620-65 EWI(m)/EWA(d)/T/EWP(t)/EWP(b)/EWA(c) IJP(c) JD/WB ACCESSION NR: AP5002347 S/0126/64/018/006/0895/0903

AUTHOR: Natapov, B. S.

TITLE: Effect of surface activity of alloying elements on the formation of secondary precipitations upon decomposition of austenite

SOURCE: Fizika metallov i metallovedeniye, v. 18, no. 6, 1964, 895-903

TOPIC TAGS: austenite, carbide crystallization, ferrite crystallization, alloying element, surface tension, alloying element solubility, grain boundary

ABSTRACT: The effect of the adsorption activity of certain alloying elements on the form of the carbide and ferrite separations along the grain boundaries of the basic phase was studied. The ratio of the surface tension of the boundaries of the different and similar phases was determined by indirect measurement of the dihedral angle of the triple junction of the grains. The value of these ratios compared with the solubilities of the alloying elements in the homogeneous phases (austenite and ferrite). Based on this, the adsorption activity of these elements was

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L 36620-65

ACCESSION NR: AP5002347

10

evaluated in comparison to the austenite-austenite and ferrite-ferrite boundaries. The elements BY O' FOWN MON V TIVAL Curand S. which are soluble in austenite, increased the floor ratio and simultaneously promoted the rupture of the carbide lattice along the austenite grain boundary, coalescence and some spheroidization of excess carbide separations at the triple junction of the grains. This confirmed the presence of the determined positive activity of these elements with respect to the austenite-austenite boundary. O. W. Mo. Ni, Mn. Cu. Ti and P, which are not significantly soluble in ferrite, increased the 6 acat / aaratio which led to the formation of structurally free carbide separations (partially spheroidized) on the ferrite grain boundaries. This confirmed the positive activi ty of these elements with respect to the ferrite-ferrite boundary. The elements Pt, Ni, Mn, and to a lesser extent Cr, are negatively active with respect to the austerite boundaries since they have greater solubility in the Y-phase and decrease the 6 carb / 6 a ratio. Si, V. Al and Cr. which are readily soluble in the 6 phase and decrease the 6 carb 6 are negative with respect to the ferrite foundaries. Negative adsorbents protect or form continuous carbide separations on the austenite-austenite and ferrite-ferrice boundaries. Orig. art. has: 3 equations and 3 tables Card 2/3

ACCESSION NR: AP5(102347 ASSOCIATION: Zaporozhak (Zaporozh Machine Construc	tv mastilia resident	y Institut im, V. Ya. Chubs	/ irya
SUBMITTED: 22Jan64	ENCL: 00	SUB CODE: MM	
NR REF SOV: 020	OTHER: 008		

11379-65 - EHT(m)/EHP(w)/EH	形(t)/EMP(b) ASI(m)=3 14JW/JD/JT S/0133/64/000/007/0642/0645	
CCESSION NR: AP4041870		
	Vasilenko, G. I., Natapov, B. S., Orekhov, G. N., V. P.	· Martin
ITLE: Case-hardening and hea	ab-treatable steels D1-2, D2-3, D1-3A (EP176) and D1-4	
OPIC TAGS: steel, case hard	ening steel, heat treatable steel, PI steel, low nickel steel, ing, steel mechanical property	
BSTRACT: The authors devel a terms of their physical and a tool currently used for high-si	oped a group of low-nickel case-hardening steels which, nechanical properties, are comparable to the high-nickel tress pieces in the machine-building industry and which it case-hardened layer in finished items. The low-nickel	
and manager ties of the	A THE MORE OF THE MAN AND MAN AND THE PARTY OF THE PARTY	
onlimal properties of Mi	d DI-4 (18KhCSN2MA) were developed to replace steels while steel DI-2A or EP176 was designed to replace steels white steel DI-2A or EP176 was designed to replace steels white steel DI-2A or EP176 was designed to replace steels	i

	AP4041870		
han for the old. clusively as a recompletely or pa of the new types, available in large the proper prope	Steel DI-3, which does not contain a placement for type 12KhN3A steel. 'rtially substituted by tungsten in the provision was inade to use the chrose quantities throughout the country. rtions of elements which promote and the low-carbon alloys, a study was a	The molybdenim in DI-3 steel may be ratio Mo: W = 1:3. In the development mium-nickel-molybdenum steel scraps Particular attention was directed at d impede case-hardening. For this purpose of the mutual effect of the basic e carbon concentration in the layer.	980X
it was discovered imits (upper and a table. No more permitted in the thoroughly tested	d that the greatest effect is exerted by I lower) of the content of the basic elements of the basic elements of the basic elements and the content of the hardening in the content of t	ements in the new steels are shown in than 0.03% sulfur and phosphorus is ited properties of the steels were emperature in the 800 - 950C range on oth DI-2 and DI-3 showed high strength	××

L 11379-65

ACCESSION NR: AP4041870

tempering temperature on the mechanical properties of type DI-2 steel was also studyles and high tempering was recommended in an interval of 536 - 600C. It was further recommended that steel DI-2 be used for air hardening in a disk to 80 mm, and with oil hardening to 150 - 200 mm. Steel DI-3A and DI-4A are recommended for sections to 80 mm, and steel DI-3 - to 40 mm. The effect of long-term high-temperature heating on the new types was found to be negligible. These steels are distinguished by fine grain, the size of which, on heating to 1,000C, remains within 7-6 units. In terms of resilience (impact ductility), the new steels are comparable to high-alloy steels and retain rather good impact toughness even at a temperature of -196C. The article indicates that the new steels are highly resistant to notelling (incising). For case-hardened items which operate under conditions of variable loads, an important characteristic is the endurance limit, which for these new economical steels is equal to that of high-nickel steels. A layer-by-layer chemical analysis showed that the carbon saturation of the case-hardened layer and its depth are the same in the new steels as in the high-alloy steels, but that the content of residual austenite is smaller. A further advantage of the new steels is the higher weakening temperature during tempering, which makes it possible to recommend them for items designed to function at temperatures up to 250 * 300C. The new low-cost steels also lend themselves well to nitriding. "Y. Ye. Pronin, G. Kh. Galanyev, Yu. P. Shamil', T. M. Babkov,

Cara3/4

L 11379-65 ACCE8SION NR; AP404187	o la	1 / 编	
L. I. Yefremova, I. P. Ba	nas, M. S. Kunin, G. V. Kuly*gin, Ye		
L. G. Kozy reva, S. 2. 11 and T. V. Levchenko also	ook part in the work." Orlg. art. has	: 1 table.	
ASSOCIATION: none		ENCL: 00	
Submitted: 00	DATE SEL: 30Jul64		
SUB CODE: MM	NO REF SOV: 002	OTHER: 000	
ard 4/4			

L 31128-65 EWT(m)/EHA(d)/EMP(t)/EWP(b) Pad IJP(c) JD/HW/JC

ACCESSION NR: AP5002940

S/0129/65/000/001/0012/0015

28 26

AUTHOR: Natapov, B.S.; Ol'spane skly, V. Ye.; Ponomarenko, Ye. P.

TITLE: Influence of the alloying elements on the shape of secondary formations in nickel-based heat resistant alloys μ

SOURCE: Metallovedeniye i termicheskaya obrabotka metallov, no. 1, 1965, 12-15

TOPIC TAGS: nickel alloy, secondary formation, heat resistant alloy, excessive phase, matrix grain, alloy additive, alloying element, alloy mechanical property

ABSTRACT: To improve the mechanical properties and resistance of heat resistant alloys, it is important to consider the formation of excess phase at the matrix grain boundary. The different shapes of these formations (thin films, continuous or interrupted network, spheroidal coagulations), as well as the abuttment angles (two-face angle 0) of three neighboring particles and the accumulation of free boundary energy are discussed. The purpose of the present work was to determine the role of some alloying elements in the formation of intergrain boundaries. For this purpose these elements were introduced into a heat resistant alloy of the following composition: 0.15-020% C, 15-20% Fe, 14-16% Cr. 2.5-3.5% Mo. 2.9-3.5% W, the balance being Ni. The additives consisted of Ti, Al. 2 B, Nb, Zr and mixed metals (Ce+Nd+Pr). The influence of each of these additives is

Cord 1/27 & Misch Metal

L 31128-65			2
ACCESSION NR: AP5	002940 The conclusion is that the		
phase in small sphero of the alloy, thereby I has: 1 figure, 3 form	The conclusion is that the B. Zr and mixed metal roidal particles that it enhalmproving the operational rules, and 2 tables.	reliability of the proc	luct. Orig. art.
building institute)			
		SUB COULE	MMC
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NATAPOV, B.S.; VASILENKO, G.I.; OL'SHAMETSKIY, V.Ye.

Character of carbide phase distribution in the carburized layer of alloy steels. Izv. vys. scheb. zav.; chern. met. F no.2:134-741 (65. (MIRA 18:2))

1. Zaporozhskiy mashimostroitel'nyy institut.

OL'SHANETSKIY, V.Ye.; NATAPOV, B.S.

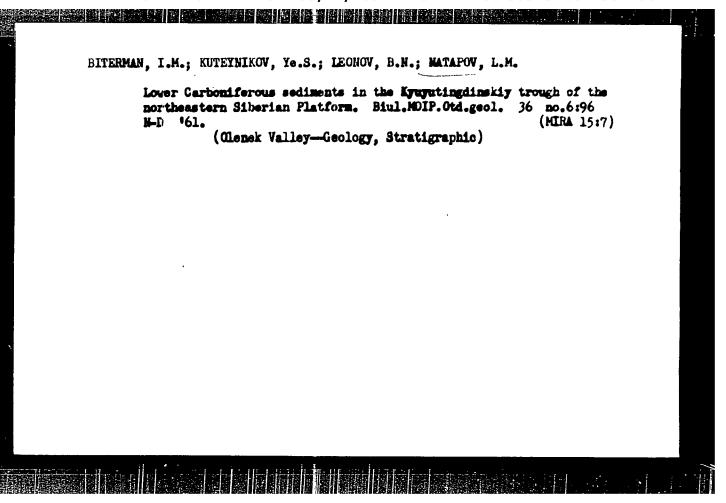
Evaluating the amount of anomalous structure in steels. Izv. vys. ucheb. zav.; chern. met. 8 no.9:158-162 '65.

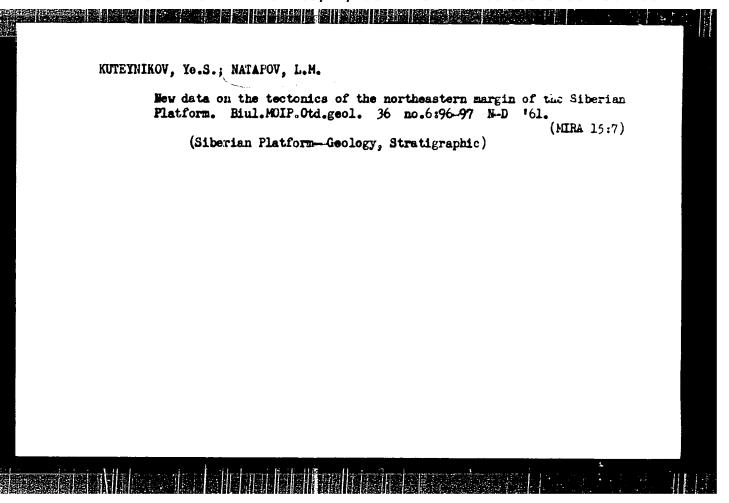
(MIRA 18:9)

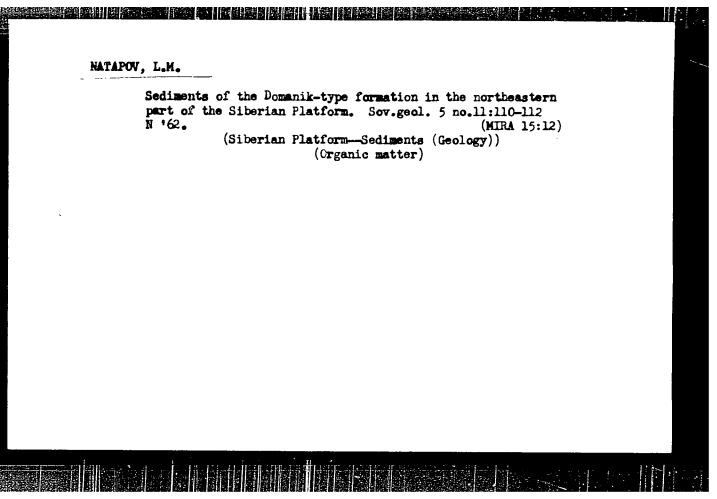
1. Zaporoshskiy mashinostroitel'nyy institut.

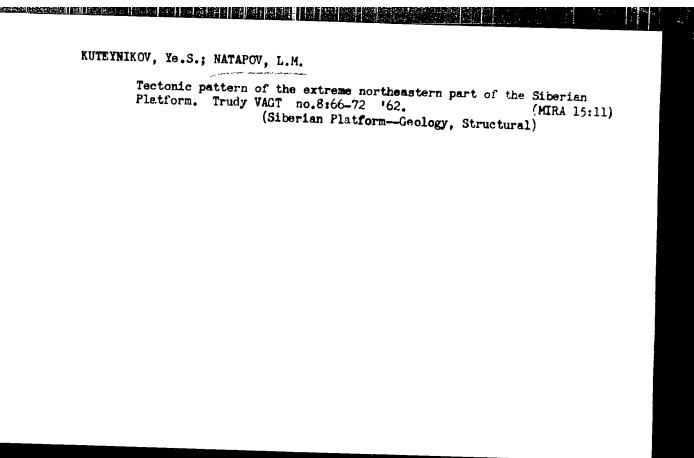
L 29383-66 EWT(m)/EWP(t)/ETI IJP(c) JD/HW/WB ACC NR: AP6016586 (N) SOURCE CORE
(a) (a) (b) (a)
AUTHOR: Gayduk, V. V.; Koval', A. D.; Natapov, B. S.
ORG: 7200001 W. 14
ORG: Zaporozh Machine-Building Institute (Zaporozhskiy mashinostroitel'-
TITLE: The structure and properties of heat-resistant ZhS-type nickel
SOURCE: Metallovedeniye i termicheskaya obrabotka metallov, no. 5,
1966, 20-22 and insert facing p. 33
TOPIC TAGS: mickel allow
TOPIC TAGS: nickel alloy, heat resistant alloy, alloy heat treatment, alloy rupture life, alloy structure, chromium containing alloy, tung-
sten containing allow tunger
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ABSTRACT: The effect of annealing temperature on the structure and
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1 CARALUGA The allow amany
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the stress of
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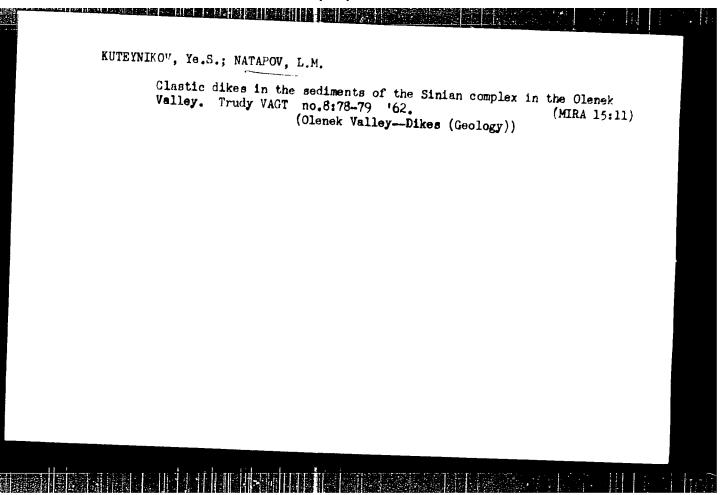
the annealing has to be performed in a protection the same of the annealing has to be performed in a protection that the same is a figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath, or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath or at lower temperature. Orig. art. has: 3 figures. [ww] salt bath	oughness, 6.3 f properties he danger of	mkg/cm ² . The most were achieved by and oxidation makes it to wances: Anot less the	nealing at 12000 r necessary, however han 0.3 mm. At sm	to use greater aller allowances tmosphere, in a	
	hath. OX	at love :			•
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KUTEYNIKOV, Ye.S.; NATAPOV, L.M.

Interpretation of fracture traces on black-and-white aerial photographs as revealed by the studies of the northeastern margin of the Paleozotc area in the Siberian Platform. Trudy VACT no.8:

130-136 '62.

(Siberian Platform-Faults (Geology))

(Siberian Platform-Aerial photogrammetry)

BITERMAN, I.M.; KUTEYNIKOV, Yo.S.; LEONOV, B.N.; NATAPOV, L.M.

New data on the lower Carboniferous deposits of the northeastern part of the Siberian Platform. Dokl.AN SSSR 144 no.3:613-616
My 162. (MIRA 15:5)

1. Vsesoyuznyy aerogeologicheskiy trest. Predstavleno akademikom A.L.Yanshinym.

(Siberian Platform—Geology, Stratigraphic)

KUTEYNIKOV, Ye.S.; NATAPOV, L.M.

New data on the tectonics of the northeastern edge of the Siberian

1. Vsesoyuznyy aerologicheskiy trest. Predstavleno akademikom D.I.Shcherbakovym.

(Siberian Platform-Geology, Structural)

Platform. Dokl. AN SSSR 149 no. 6:1405-1408 Ap '63. (MIRA 16:7)

Characteristics of the tectonic development of the Glenek highland in the Late Pre-Cambrian. Dokl. AN SSSR 101 no.5:1173-1176 Ap '65.

1. Submitted February 15, 1964.

FACC NR. AP6034490 SOURCE CODE: UR/0210/66/000/006/0050/0059

AUTHOR: Zonenshayn, L. P.; Natapov, L. M.; Uflyand, A.K.

ORG: All-Union Aerogeological Trust/(Vsesoyuzhyy aerogeologicheskiy

TITLE: Structure of the Aldan branch of the Priverkhoyansk foredeep

SOURCE: Geologiya i geofizika, no. 6, 1966, 50-59

TOPIC TAGS: geologic exploration, anticline, geologic surveyence, foredeep, at a transfer fectories

ABSTRACT: The structure of the Aldan branch of the Priverkhoyansk foredeep is described. Steep flexures alternating with gently sloping echelon brachysynclines characterize the boundary region between the Verkhoyansk folded region and the Priverkhoyansk foredeep. The limiting folds are oriented at a steep angle to the foredeep strike, plunging east-southeast. These folds can be traced within the inner zone of the foredeep. A system of narrow anticlines, separated by broad synclines is also found in the inner zone of the foredeep. The entire Verkhoyansk complex consists of Permian-Cretaceous formations. The outer zone of the foredeep is composed of Jurassic and Cretaceous formations superposed on a Lower Paleozoic basement. The inner and

Card 1/2

NDC: 653.98: 651.70+851.24(571.56)

ACC NR: AP6034490

outer zones are separated by a marginal suture-type deep-seated fault.

Orig. art. has: 1 figure

SUB CODE: 08/ SUBM DATE: 23Apr64/ ORIG REF: 012/ OTH REF: 006

PARUSNIKOV, V.H., NaTAROVA, R.I.; MALIKOVA, L.P.

Cleaning of hot-drawn tantalum wire from graphite -perication.
TSvet. met. 36 no.10:73-76 0 '63.

(MIRA 16:12)

ACCESSION NR: AP4011290

S/0136/64/000/001/0066/0069

AUTHOR: Natapoya, R. I.; Kirsanova, T. A.; Malikova, L. P.; Sokolov, Yu. A.; Parusnikov, V. N.

TITLE: Cold drawing of tantalum wire

SOURCE: Tsvetny*ye metally*, no. 1, 1964, 66-69

TOPIC TAGS: tantalum wire, tantalum wire drawing, tantalum copper plating, cold drawing, wire drawing, copper plated tantalum wire

ABSTRACT: A method for smooth drawing of tantalum wires (Authors certificate Nr. 148373) was devised to eliminate wire rupture and gas absorption by the metallic wires which cause the wire to possess poor mechanical properties. Since the use of ordinary lubricants and oxidizing of the metal surface does not eliminate these difficulties, it is proposed that the tantalum material after cleaning be copper plated by hot dipping in an inner atmosphere. Hot-drawn wire was cleaned of aquadag and oxides by electrolytic etching. Hot copper plating of

Card 1/2

ACCESSION NR: AP4011290

the cleaned wire was done in argon by drawing the wire through a graphite crucible with molten copper. Rate of drawing and temperature must be strictly controlled for uniform coating. The latter is uniformly deformed during cold drawing and does not peal off. Cold drawing of 100-200 micron diam. wire(coating 1-2 microns) to a maximum size of 40-60 microndiam. can be achieved. For drawing to finer wires electrolytic copper plating should be superimposed thereon (100-200 micron diam primary wire 10-20 micron diam final wire, 3-5 micron coating achieved in two passages at a rate of 1.5-2 m/min, 20sec. in the bath, 20 amp/sq. in. current density). Electrolytic coating should be applied over etched hot coating for better uniformity and smoother drawing of small gauge wires. After drawing, coating should be electrolytically or chemically removed. Thus, perfect cold drawing of finest gauges becomes possible due to copper plating. Rate of drawing ranges from 20-15 m/min for 30-250 micron diam to 8-2 m/min- for 10-30 micron diam. Orig. art has: 3 figures.

ASSOCIATION: None

SUBMITTED: 00

DATE ACQ: 14Feb64

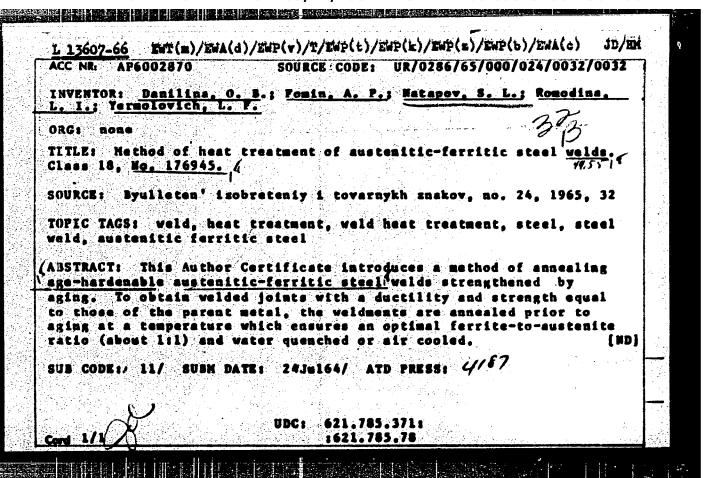
ENCL: 00

SUB CODE: EL

NO REF SOV: 005

OTHER: 002

Card 2/2



KONONENKO, G.I., inzh.; NATARIUS, N.A., inzh.

Loading and unloading equipment and conveying machinery made
in England. Mekh.i avtom.proisv. 15 no.11:56-61 N '61.

(Great Britain—Loading and unloading—Equipment and supplies)

(Great Britain—Conveying machinery)

(Moscow—Exhibitions)

APPROVED FOR RELEASE: 03/14/2001 CIA-RDP86-00513R001136120012-4"

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ABDULIN, A.; ALEKSEYEV, I.; BANTLE, O.; BOBROV, L.; BOZHANOV, B.;
BOYKO, V., BONDAREV, K.; BORZOV, V.; VERKHOVSKIY, N.; GUBAREV, V.;
GUSHCHEV, S.; DEBABOV, V.; DIKS, R.; DMITRIYEV, A.; ZHIGAREV, A.;
ZEL'DOVICH, Ya., ZUBKOV, B., IRININ, A.; IORDANSKIY, A.;
KITAYGORODSKIY, P.; KLYUYEV, Ye., KLYACHKO, V.; KOVALEVSKIY, V.,
KNORRE, Ye.; KONSTANTINOVSKIY, M.; LADIN, V.; LITVIN.SEDOY, M.;
MALEVANCHIK, B.; MANICHEV, G.; MEDVEDEV, Yu.; MEL'NIKOV I.,
MUSLIN, Ye.; ATTARIUS Ya.; REYFAKH, A., NIKOLAYEV, G.; NOVOMEYSKIY, A.,
OLISHANSKIY, N.; OS'NIN, S.; PODOL'NYY, R.; RAKHANOV, N.; REPIN. L.;
RESHETOV, Yu.; RYBCHINSKIY, Yu., SVOREN', R.; SIFOROV, V.; SOKOL'SKIY, A.;
SPITSYN, V.; TEREKHOV, V.; TEPLOV, L.; KHAN'KOVSKIY, A.; CHERNYAYEV, I.;
SHAROL', L.; SHIBANOV, A.; SHIBMEV, V.; SHUYKIN, N.; SHCHUKIN, O.;
EL'SHANSKIY, I.; YUR'YEV, A.; IVANOV, N.; LIVANOV, A.; FEDCHENKO, V.,
DANIN, D., red.

[Eureka] Evrika. Moskva, Molodala gvardila, 1964. 278 p.
(MIRA 18:3)
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NATARIUS, YA I.

AUTHOR:

Natarius, Ya.I., Engineer

98-7-16/20

TITLE:

Research Work in the Field of Concrete (Issledovatel'skiye

raboty v oblasti betona)

PERIODICAL:

Gidrotekhnicheskoye Stroitel'stvo, 1957, No 7, pp 52-53 (USSR)

ABSTRACT:

Extensive research work was conducted at the Glockner-Kaprun construction project regarding water permeability and frost resistance of concrete. It was found that frost resistance increased when dust particles were removed from the filling materials. The applied cycles of freezing and thawing varied from 8 to 12, and 16 hours (the thawing period being 4 hours) at temperature readings of from -20 to +15° C. Tests were conducted with samples of varying granular structures of the filling materials, and with 270 kg of cement per cu m of concrete. The percentage of cement had little influence on the resistance to freezing, whereas the quality of the cement influenced the degree of frost resistance. The selection of plasticizers is determined by the necessity to reduce the water content of the mixture. Frost resistance of concrete is also increased when fine sand is not used in the event of the application of plasticizing additives with air attracting properties.

Card 1/2

Research Work in the Field of Concrete

98-7-16/20

As a result of the conducted tests, concrete with a percentage of 125 kg of cement per cu m of concrete was obtained, suitable for massive structures, whereas concrete with a content of 250 kg/cu m may be considered to be satisfactorily frost resistant as surface sections of concrete spill-dams. Concrete mixtures with contents of 225 kg/cu m showed satisfactory resistance to permeability at pressures from 1.5 to 10 atmospheres exerted during 24 hours, and 15 atmospheres during 7 days. The removal of dust from the sand was accomplished with the hydraulic classifier "Reaks". This article contains 1 figure, 2 tables, 4 diagrams, and 3 references, 2 of which are in English and one in German.

AVAILABLE:

Library of Congress

Card 2/2

AUTHOR: Natarius, Ya.I., Engineer SOV-98-58-9-18/21

TITLE: The Glen Canyon Hydro-Electric Installations on the Colo-

rado River (Gidrouzel Glen Ken'on na r.Kolorado)

PERIODICAL: Gidrotekhnicheskoye stroitel'stvo, 1958, Nr 9, pp 50 - 52

(USSR)

ABSTRACT: The main characteristics of hydroelectric installations

in Clen Canyon on the Colorado river are given. There are 3 diagrams and 3 references, one of which is Soviet

and 2 American.

1. Hydroelectric power systems

Card 1/1

HATARIUS, Ym.I.,inzh.

Salime(Spain) Hydroelectric Power Station. Gidr.stroi. 27 no.3:53-55
Mr '58.

(MIRA 11:4)

(Salime hydroelectric power station)

HATARIUS, Ta.I., insh.

Glen Canyon hydroelectric project on the Colorado River. Gidr.stroi.
27 no.9:50-52 S '58. (MIRA 1::21)

(Colorado River---Power utilization)

15(6) SOV/98-59-3-14/17 AUTHOR: Natarius, Ya.I., Engineer

TITLE: The Utilization of Gamma Radiography for the Examin-

ation of Concrete (Primeneniye gamma-radiografii

pri issledovanii betona)

Gidrotekhnicheskoye stroitel'stvo, 1959, Nr 3, pp PERIODICAL:

55-57 (USSR)

ABSTRACT: The author describes a method of controlling the

quality of reinforced concrete parts by utilizing gamma rays. This method, used abroad, is described from foreign sources. There are 3 references, 1 of which is English, 1 Indian and 1 Soviet.

Card 1/1

sov/98-59-8-15/33

10(0) AUTHOR:

Natarius, Ya.I., Engineer

TITLE:

The Organization of Scientific Research Work in the Field of Hydraulics in Australia

PERIODICAL:

Gidrotekhnicheskoye stroitel'stvo, 1959, Nr 8, pp 52-53 (USSR)

ABSTRACT:

The article is a discussion of the work carried out by the Special Board for the Exploitation of the Water Resources of the Snowy Mountains, within the limits of which flow 3 of the largest rivers of the Australian continent, the Snowy, Murray and Hume. There is 1 Australian reference.

Card 1/1

NATARIUS, Ya.I., ingh.

Using the method of gemma-radiography in testing concrete. Gidr.
stroi. 28 no.3:55-56 Mr '59.

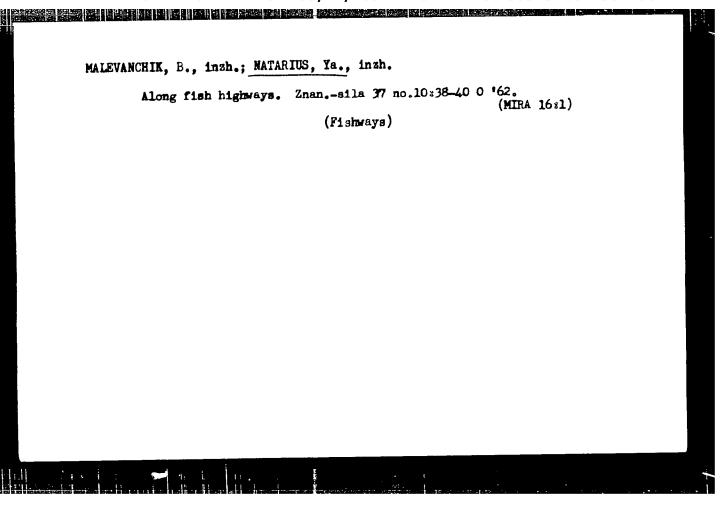
(Concrete--Testing)

(Radioisotopes--Industrial applications)

NATARIUS, Ya.I., inzh.

Construction and operation of fishways. Gidr. stroi. 33 no.11:
50-52 N '62.

(Fishways)



EUMSHTEYN, S.I.; NATAROV, A.I.; MIKHAYLOV, K.I., red.

[Manual for the driver of the second class; construction, maintenance and operation of motor vehicles] Uchebnoe posobie shofera vtorogo klassa; ustroistvo, tekhnicheskoe obsluzhivanie i ekspluatatsiia avtomobilei. Moskva, DOSAAF, 1965. 495 p.

(MIRA 18:5)

NATAROV, Boris Fedorgyich; CHRKHOVIY, M., veduchiy redsktor; PATSALYUK,

F., textinicheskiy redsktor

[Wide-screen soution pictures] Shyrokoekranne kino. Kyiv,
Dersh. vyd-vo tekha. lit-ry UBSR, 1956. 36 p. (MLRA 10:5)

(Motion-picture projection)

MATAHOV, B.F., kand.tekhn.nauk. dots.

Three-dimensional motion pictures. Hauka i shyttia & no.5:
14-17 My '58.

(MIRA 13:4)

(Motion pictures, Three-dimensional)

NATAROV, B.F.

Quality of Stereophonic sound in motion-picture theaters. Tech.
kino i telev. 4 no.6:45-48 Je '60. (MIRA 13:7)

1. Kiyevskiy politekhnicheskiy institut.
(Stereophonic sound systems)
(Motion-picture theaters)

Mumber of channels in a stereophonic sound reproducing system in wide-screen motion-picture theaters. Tekh.kino i telev. 4 no.7: 49-51 Jl '60. (MIRA 13:7) 1. Kiyevskiy politekhnicheskiy institut. (Motion-picture theaters) (Stereophonic sound systems)

ACC NR. AT6034455

(N)

SOURCE CODE: UR/0000/66/000/000/0194/0200

AUTHOR: Ol'shanetskiy, V. Ye.; Materov, B. S.

ORG: none

TITIE: Effect of the absorptive activity of alloying elements on the nature of the distribution of excess phases along the grain boundaries of nickel base alloys

SOURCE: AN SSSR. Institut metallurgii. Svoystva i primeneniye sharoprochnykh splavov (Properties and application of heat resistant alloys). Moscow, Isd-vo Nauka.

TOPIC TAGS: nickel base alloy, phase diagram, metal grain structure

ABSTRACT: The relative change in free energy in the transition from like grain boundaries to unlike grain boundaries is determined by the expression

$$\frac{a^{ij}}{a^{ii}} = \frac{1}{2\cos\frac{3^{ij}}{2}},\tag{1}$$

where β^{-ij} is the dihedral angle in the triple junction of two grains of the matrix and a grain of the excess phase. Then, for segregated phases with a lattice character, the following inequalities are valid

$$0 < \beta^{ij} < 60^{\circ}_{ant} \frac{1}{2} < \frac{\sigma^{ij}}{\sigma^{ij}} < \frac{1}{V_3}$$

Card 1/2

ACC NR: AT6034455

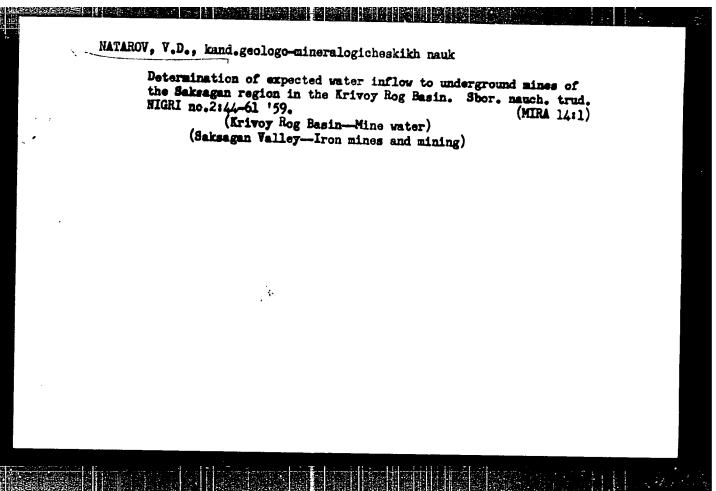
and for segregated phases of spherical form

$$\beta^{ij} > 60^{\circ} = \frac{\sigma^{ij}}{\sigma^{ii}} > \frac{1}{V\bar{3}}$$
.

To clarify the role of certain alloying elements in the change in the energy state of the grain boundaries in heat resistant nickel base alloys, these elements were introduced into an alloy with the following initial chemical composition: 0.15-0.20% carbon; 15-20% iron; 14-16% chromium; 2.8-3.5 molybdenum; 2.9-3.5% tungsten; remainder nickel. The following alloying elements were investigated: Ce, Pr, Nd, B, Zr, Al, Nb, and Ti. The effect of the added elements is shown in tabular form and by microphotographs of the alloys. In general, it is concluded that introduction into heat resistant nickel base alloys of small amounts of alloying elements, including boron and zirconium, brings about a favorable form of distribution of the excess phases, which should promote strengthening of the intergrain boundaries in these alloys. Orig. art. has: 5 formulas, 1 figure and 1 table.

SUB CODE: 11/ SUBM DATE: 10 Jun 66/ ORIG REF: 012/ OTH REF: 005

Card 2/2



MATAROV, V.D.; KAZAK, V.M.

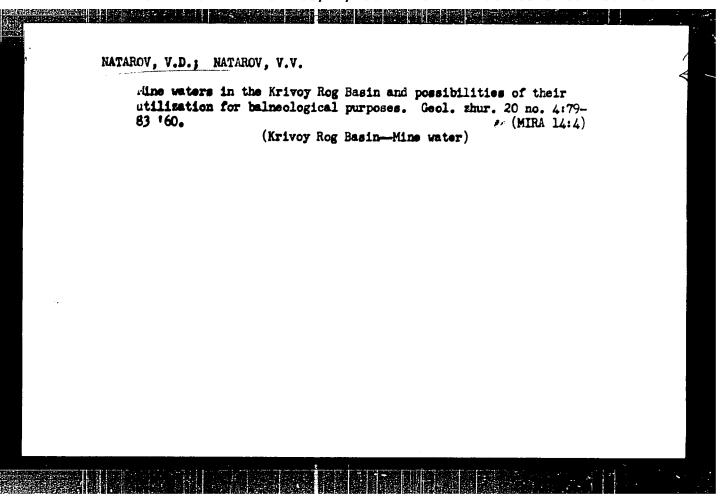
Method of determining intersticial water resources in the Krivov Rog Basin. Sbor. nauch. trud. WIGRI no.2:62-79 '59. (MIRA 14:1)

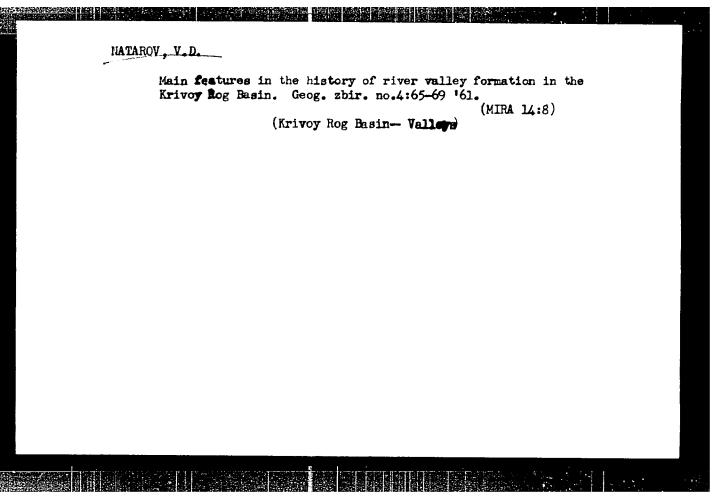
(Krivov Rog Basin-Water, Underground)

Method of determining anticipated inflows into mine openings and open pits with a calculation of drainage time under conditions of the Krivoy Rog Basin. Sbor. nauch. trud. NIGRI no.7:60-69 (MIRA 14:12)

(Krivoy Rog Basin—Mine drainage)

(Water, Underground)





NATAROV, V.D. Karsts and karst waters in Pre-Cambrian rocks in the Saksagan' region of the Krivoy Rog Basin. Sov.geol.: 4 no.9:143-148 5 '61. (MRRA 14:11) 1. Krivorozhskiy nauchno-issledovatel'skiy gornorudnyy institut. (Krivoy Rog Basin--Karst)

MATAROV, V.D.; MALAKHOV, N.L.

Manganese ores in the Krivoy Rog Masin. Razved. 1 okh. nedr. 27 no.3:
4-6 Mr '61. (MIRA 14:5)

1. Mauchno-issledovatel'skiy gornorudnyy institut (for Natarov).
2. Rudnik Ingulets, Krivoroznakiy shelezorudnyy basseyn (for Malakhov).

(Krivoy Rog Basin—Manganese ores)

Role of underground waters in the formation of high-grade iron ores in the Krivoy Rog iron-ore basin, Geol. shur. 23 no.2:53-58 '63. (MIRA 16:6) 1. Krivoroshskiy nauchno-issledovatel skiy gornorudnyy institut. (Krivoy Rog.—Mater, Underground) (Krivoy Rog.—Iron ores)

BRYZGALOVA, Ye.V., kand.ekon.nauk; MATAROV, V.F., inzh.-ekonomiat

Biffect of the utilization of waste and by-products on the economy of the shales-gas industry. Trudy LIEI no.25:84-96
159.

(Kohtla-Jdrva--Oil shales)

(Kohtla-Jdrva--Oil shales)

